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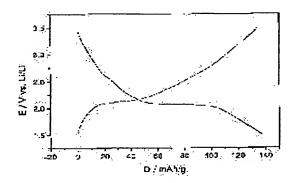
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(54) POSITIVE ACTIVE MATERIAL FOR NONAQUEOUS ELECTROLYTE SECONDARY BATTERY

(57)Abstract:

PROBLEM TO BE SOLVED: To provide an ironcontaining positive electrode active substance for a nonaqueous electrolyte secondary battery which is of low cost and yet with large charge and discharge capacity and superb charge and discharge cycle characteristics.

SOLUTION: As a positive active material, an alkali metal-containing iron oxide with hexagonal rock salt structure of which a part of the surface is clad with vanadium oxide. Further, it is preferable to use α-NaFeO2 as an alkali metal-containing iron oxide.



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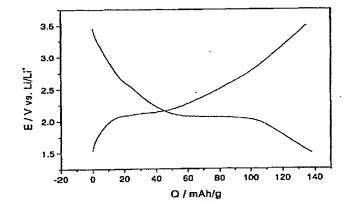
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(54) 【発明の名称】 非水電解質二次電池用正極活物質

(57)【要約】

【 課題】安価で、しかも高い充放電容量を持ち、良好な 充放電サイクル特性を持つ非水電解質二次用鉄含有正極 活物質を提供する。

【解決手段】正極活物質として、六方晶層状岩塩型構造を有するアルカリ金属含有鉄酸化物の表面の少なくとも一部にバナジウム酸化物が担持されるのを使用する。なお、上記アルカリ金属含有鉄酸化物としては、α-Na Fe O2を用いるのが好適である。



【 特許請求の範囲】

【請求項1】六方晶層状岩塩型構造を有するアルカリ金 属含有鉄酸化物とバナジウム酸化物とが混合され、前記 アルカリ金属含有鉄酸化物の表面の少なくとも一部に前 記バナジウム酸化物が担持されていることを特徴とする。 非水電解質二次電池用正極活物質。

【請求項2】前記アルカリ金属含有鉄酸化物がα-Na Fe O2であることを特徴とする非水電解質二次電池用 正極活物質。

【発明の詳細な説明】

[0001]

[発明の属する技術分野] 本発明は、非水電解質二次電池用正極活物質に関する。

[0002]

【 従来の技術】近年、携帯用電話、ビデオカメラ等の小型電源および電気自動車、電力平準化用の大型電源として、高エネルギー密度、高出力密度を有する二次電池、特にリチウム二次電池が大きな注目を受けている。このリチウム二次電池に用いられる材料として、正極にはリチウム遷移金属酸化物が、負極には黒鉛、低温焼成炭素、酸化物、リチウム合金およびリチウム金属が提案されている。

【 0 0 0 3 】現在、正極活物質として使われているコバルト酸リチウム(Li CoO2) は高価であり、将来予測されるリチウム二次電池の大量消費に対応するためには、より安価で埋蔵量が豊富な正極活物質の開発が重要である。現在、マンガンやニッケル、鉄を含む酸化物がリチウム二次電池用正極活物質として精力的に研究されている。中でも鉄は最も安価で環境負荷の小さい材料であるため、鉄を主体として含む酸化物は次世代リチウム二次電池用正極活物質として大変魅力的である。

【 0 0 0 4 】鉄を主体として含むリチウム二次電池用正極活物質として、これまで種々のリチウム含有鉄酸化物が提案されてきた。例えば、トンネル構造または層状ジグザグ構造を有するリチウム鉄複合酸化物(Li Fe O 2)(例えばJ. Electrochem. Soc., 143, 2435(1996))、オリビン型Li Fe PO4(J. Electrochem. Soc., 144, 1609(1997))、さらに、六方晶層状岩塩型構造を有するLi Fe O2(例えば特開平10-67519)等が挙げられる。

[0005]

【 発明が解決しようとする課題】上記トンネル構造または層状ジグザグ構造を有するLi Fe O2は、初期にLi Co O2を凌ぐ高い充放電容量(150 mAh/g以上)を有するが、10 サイクルの寿命試験で放電容量が初期容量の80%以下に低下し、充放電サイクル特性が低い問題点がある。オリビン型Li Fe P O4の放電容量は140 mAh/g 以下であり、電池活物質として不十分である。一方、六方晶層状岩塩型構造を有するLi

Fe O2は、J. Electrochem. Soc., 144, L177(1997)で示されているように、 充放電容量が極めて低く(10 mAh /g以下)、さらに た放電サイクル特性が低い課題がある。

【 0 0 0 6 】従って、これまで1 5 0 mAh /g 以上の高い放電容量を有し、なおかつ充放電サイクル特性に優れたリチウム鉄複合酸化物は得られていない。

【 0007】一方、アルカリ電池正極活物質であるMnO2の利用率向上に、Bi 2O3およびPbO等の金属酸化物を混合することが有効であることが知られている(H.S.Wroblowaand N.Gupta, J.Electroanal.Chem., 238, 93(1987))。金属酸化物の作用として、近年、触媒性が指摘されている(De Yan G Qu,

J. Appl. Electrochem., 29,51 1(1999))。マンガンにはBi 2O3およびPbO との混合が有効であるのに対し、他の遷移金属、例えば 鉄にいかなる金属酸化物が良好な作用をもたらすかは明 らかになっていない。

【 0 0 0 8 】本発明はかかる金属酸化物の添加効果に注 目したものであり、その目的とするところは、高い充放 電容量を持ち、さらに良好な充放電サイクル特性を持つ 非水電解質二次用鉄含有正極活物質を提供することにあ る。

[0009]

【 課題を解決するための手段】本発明の非水電解質二次 電池用正極活物質は、六方晶層状岩塩型構造を有するア ルカリ 金属含有鉄酸化物とバナジウム酸化物とが混合さ れ、前記アルカリ 金属含有鉄酸化物の表面の少なくとも 一部に前記バナジウム酸化物が担持されていることを特 徴とする。

【 $0\ 0\ 1\ 0$ 】 さらに、本発明の非水電解質二次電池用正極活物質では、前記アルカリ 金属含有鉄酸化物が $\alpha-N$ a F e O2であることを特徴とする。

[0011]

【 発明の実施の形態】本発明では、六方晶層状岩塩型構造を有するアルカリ金属含有鉄酸化物の表面の少なくとも一部にバナジウム酸化物を担持させることにより、アルカリ金属含有鉄酸化物の非水電解質二次電池用正極活物質としての利用率を大幅に増加させ、サイクル特性を良好にすることができる。

【 0 0 1 2 】 本発明のアルカリ 金属含有鉄酸化物には、 Li Fe O2、a-Na Fe O2を用いることができ、N a を一部Li で置換したLi 1-xNa xFe O2(0 <x <1)、および鉄の一部を他の遷移金属元素で置換した Li M1-yFe yO2(M=Co, Mn, Ni)(0 <y <1)も用いることができる。ただし、いずれも六方晶 層状岩塩型構造を有することを特徴とする。

【 0 0 1 3 】 本発明で用いられるバナジウム酸化物には、 V 2O 3、 V 2O 4、 V 2O 5を用いることが可能であ

る。

[0014]

【 実施例】以下に本発明なる非水電解質二次電池用正極 活物質の実施例を説明する。しかし、本発明は以下の実 施例に限定されるものではない。

【0015】[実施例1] 酸化第二鉄(a-Fe 2O3) および過酸化ナトリウム(Na 2O2) をそれぞれ0.03モル秤量し、乳鉢で混合し、ペレット化した後、酸素雰囲気下、550℃で20時間焼成した。つづいて、試料を粉砕し、再度550℃で20時間焼成することにより、六方晶層状岩塩型構造を有するa-NaFe O2を得た。試料の秤量、乳鉢での混合はすべてアルゴン雰囲気下のグローブボックス内で行った。

【 0016 】 つぎに、上記で得られたa-Na Fe O2と V2O5をモル比が40:1 になるようにメタノール中で 湿式混合し、80 ℃で乾燥することによって、a-Na Fe O2の表面にバナジウム酸化物が担持された正極活物質を作製した。正極活物質としてのa-Na Fe O2と V2O5の混合物75 重量部に、導電剤としてのポリフッカビニリデン(PVDF)5 重量部を加え、溶剤であるN-メチルー2 ピロリドンと湿式混合してスラリーにした。このスラリーを集電体であるアルミニウムメッシュの両面に塗付した後、1t /c m^2 で加圧成形し、真空下にて230 ℃で乾燥し、大きさ15 mm ×15 mm ×0.5 mmの本発明正極板(A1) を作製した。

【 0 0 1 7 】 [実施例2] 実施例1 に基づき得られる a -N a F e O2と V 2O5をモル比が4 0: 2 になるように混合し、正極活物質としたこと以外は実施例1 と同様にして、本発明正極板(A2) を作製した。

【 0 0 1 8 】 [実施例3] 実施例1 に基づき得られるa -Na Fe O2と V2O5をモル比が4 0:4 になるように混合し、正極活物質としたこと以外は実施例1 と同様にして、本発明正極板(A3) を作製した。

【 0 0 1 9 】[実施例4] 実施例1 に基づき得られる a -N a F e O 2と V 2O 5をモル比が4 0 : 1 0 になるように混合し、正極活物質としたこと以外は実施例1 と同様にして、本発明正極板(A 4) を作製した。

【 0 0 2 0 】 [実施例5] 実施例1 に基づき得られる a - N a F e O 2と V 2O 5をモル比が4 0 : 2 0 になるように混合し、正極活物質としたこと以外は実施例1 と同様にして、本発明正極板(A 5) を作製した。

【 0021】[実施例6] 実施例1 に基づき得られるa -Na Fe O2とV2O5をモル比が40:40になるように混合し、正極活物質としたこと以外は実施例1と同 様にして、本発明正極板(A6)を作製した。

【 0 0 2 2 】 [実施例7] 実施例1 に基づき得られる a - N a F e O 2と V 2O 5をモル比が4 0 : 6 0 になるように混合し、正極活物質としたこと以外は実施例1 と同様にして、本発明正極板(A7) を作製した。

【 0 0 2 3 】[実施例8] 実施例1 に基づき得られるa -Na Fe O2と V2O5をモル比が4 0:80 になるように混合し、正極活物質としたこと以外は実施例1と同様にして、本発明正極板(A8) を作製した。

【 0 0 2 4 】 [実施例9] 実施例1 に基づき得られるa -Na Fe O2と V2O5をモル比が4 0:100 になるように混合し、正極活物質としたこと以外は実施例1と同様にして、本発明正極板(A9)を作製した。

【 0 0 2 5 】 [実施例1 0] 実施例1 に基づき得られる a-Na Fe O2と V2O5をモル比が4 0: 120 になる ように混合し、正極活物質としたこと以外は実施例1と 同様にして、本発明正極板(A10)を作製した。

【 0 0 2 6 】[比較例1] 正極活物質としてa-Na Fe O 2を単独で用いたこと以外は実施例1と同様にして、比較正極板(B1)を作製した。

【 0027】[充放電特性] 本発明正極板(A1)~(A10) および比較正極板(B1) をそれぞれ試験極基材とし、実験用セルを構成した。対極および参照極にリチウム金属、非水電解液に1 mol /lの過塩素酸リチウムを溶解させたエチレンカーボネートとジエチルカーボネートの体積比1:1の混合溶液を用いた。

【0028】上記実験用セルを用いて、正極充放電特性 を調べた。本発明正極板(A1)~(A10)、および 比較正極板(B1) について、電流密度2 mA/g で 1.5 Vまで放電した後、折り返し2 mA/g で3.5 Vまで充電した。放電容量をC1、充電容量をC2とお くとき、充放電サイクル効率を以下の式から算出した。 充放電サイクル効率(%)=(C2/C1)×100な お、充放電試験は放電(リチウム挿入)から開始した。 【 0 0 2 9 】a-Na Fe O2と V 2O5のモル比をx、y とするとき、本発明正極板(A1)~(A10) および 比較正極板(B1)の(y/x)の値と各サイクルにお ける 充電容量との関係を表1に示す。また、本発明正極 板(A1)~(A10) および比較正極板(B1) の各 サイクルにおけるサイクル効率を表2に示す。さらに、 本発明正極板(A6)と比較正極板(B1)の3サイク ル目における充放電曲線をそれぞれ図1、図2に示す。

[0030]

【表1】

正视板		充電容量(mAh/g)		
	(y/x)	1サイクル	2サイクル	5サイクル
A 1	0.025	7	7	7
A 2	0.05	15	16	16
A 3	0.1	27	27	27
A4 .	0.25	5 2	6 5	7 1
A 5	0.5	9 2	9 6	98
A 6	1. 0	134	134	135
A 7	1. 5	175	170	168
A 8	2. 0	203	191	188
A 9	2. 5	225	208	204
A 1 0	3. 0	239	217	213
В 1		3	4	4

[0031]

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Œ	正極板		サイクル効率 (%)		
	(y/x)	1サイクル	2サイクル	5サイクル	
A 1	0.025	8 7	89	95	
A 2	0.05	8 6	94	9 7	
A 3	0.1	8 5	94	98	
A 4	0.25	8 6	98	9 9	
A 5	0.5	9 2	96	98	
A 6	1.0	8 8	96	97	
A 7	1. 5	8 6	97	98	
A 8	2. 0	8.0	99	9 9	
A 9	2. 5	7 9	98	98	
A 1 0	3. 0	7 7	98	9 9	
В 1	_	5.6	78	9 2	

【0032】図1、図2から、a-Na Fe O2とV2O5 の混合物(A6)ではa-Na Fe O2(B1)と比較 し、その充放電曲線が大きく異なることが分かる。すな わち、a-Na Fe O2(B1) では電位がなだらかに変 化したのに対し、a-Na Fe O2と V2O5の混合物(A 6) では約2 V に電位の平坦性が確認された。約2 V の 電位の平坦性はV₂O₅を活物質とした場合の初期サイク ルにおいても同様にして現れるが、以後のサイクルで は、構造がアモルファス化し、電位がなだらかに変化す ることが報告されている(小柴ら, DENKI KAG AKU, 332(1994))。また、本発明正極板に おいてV2O5のみ反応に寄与していると仮定し、初期サ イクル放電過程における容量(2.1 V~1.5 V)を 比較すると、本発明正極板(A4)では240mAh/ g となるのに対し、V2O5では約200mAh /gとな る(小柴ら, DENKI KAGAKU, 332 (19 94)).

【 0033】以上の結果、本発明正極板では、V2O5単独の特性が現れているのでなく、a-NaFeO2が充放電反応に大きく寄与していると考えられる。V2O5との混合により、比較正極板(B1)と比べて本発明正極板の容量が増加したのは、a-NaFeO2の利用率の向上に起因すると考えられる。

【 0034】Feの2価/3価の酸化還元反応は約2Vでおこることが報告されている(K. Amine et.al., J. Power. Sources, <u>81-82</u>, 221(1999))。従って、本発明正極板で

出現した約2 V の電位の平坦性(図1)は、a-NaF $e O_2$ へのリチウムの挿入・脱離にともなう $Fe O_2$ 価/3 価の酸化還元反応に起因し、 $a-NaFe O_2$ の利用率の向上によってもたらされたと考えられる。

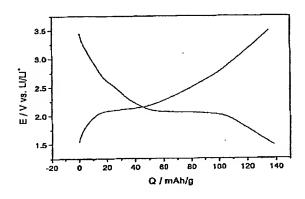
【 0035 】表1 から、(y /x)の値が1 ~3 .0 である本発明正極活物質は大きな放電容量(130 ~20 0 mAh /g)を有することが分かる。また、表2 から、(y /x)の値が0 ~2 .5 である本発明正極活物質は、比較正極板と比べてサイクル特性が優れていることが分かる。従って、(y /x)の値が1 ~2 .5 であるa-Na Fe O2と V2O5の混合物を用いることが、高容量で、良好なサイクル特性を有する非水電解質正極活物質を得る上で好ましいことが分かる。

【 0 0 3 6 】本実施例では、アルカリ 金属含有鉄酸化物にa-Na Fe O2、バナジウム酸化物にV2O5を選んで説明したが、a-Na Fe O2にV2O3、V2O4を混合した活物質や、Li Fe O2、Fe OOHにV2O3、V2O4、V2O5を混合した活物質においても同様にして、高容量で、良好なサイクル特性が得られた。

【 0 0 3 7 】本発明により、高容量で充放電特性に優れたアルカリ金属含有鉄酸化物を得ることに初めて成功し、安価で環境負荷の小さい非水電解質二次電池用正極活物質の開発に大きく寄与するものである。

[0038]

【 発明の効果】本発明の非水電解質二次電池用正極活物 質は、アルカリ 金属含有鉄酸化物の表面の少なくとも一 部にバナジウム酸化物を担持したことを特徴とし、アル 【図1】

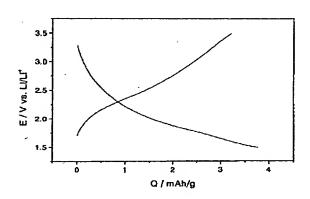


【図面の簡単な説明】

【 図1 】本発明正極板(A6) の3 サイクル目における 充放電特性を示す図である。

【 図2 】比較正極板(B1)の3サイクル目における充-放電特性を示す図である。

【図2】



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CLAIMS

[Claim(s)]

[Claim 1] Positive active material for nonaqueous electrolyte rechargeable batteries characterized by mixing the alkali-metal content ferric acid ghost and banazin san ghost which have hexagonal stratified rock salt type structure, and said banazin san ghost being supported by a part of front face [at least] of said alkali-metal content ferric acid ghost.

[Claim 2] Positive active material for nonaqueous electrolyte rechargeable batteries characterized by said alkali-metal content ferric acid ghost being alpha-NaFeO 2.

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DETAILED DESCRIPTION

[Detailed Description of the Invention]

[0001]

[Field of the Invention] This invention relates to the positive active material for nonaqueous electrolyte rechargeable batteries.

[0002]

[Description of the Prior Art] In recent years, the rechargeable battery which has a high energy consistency and high power density, especially the lithium secondary battery have received big attention as a large-scale power source for the small power source of a portable telephone, a video camera, etc. and an electric vehicle, and power equalization. As an ingredient used for this lithium secondary battery, lithium transition-metals oxide is proposed by the positive electrode, and a graphite, low-temperature baking carbon, oxide, the lithium alloy, and the lithium metal are proposed by the negative electrode.

[0003] It is expensive, and in order to correspond to mass consumption of the lithium secondary battery predicted in the future, current and the cobalt acid lithium (LiCoO2) currently used as positive active material are more cheap, and development of positive active material with abundant reserves is important for it. The oxide containing current, manganese and nickel, and iron is energetically studied as positive active material for lithium secondary batteries. Iron is the cheapest especially, and since it is the small ingredient of an environmental load, the oxide which contains iron as a subject is very attractive as positive active material for next-generation lithium secondary batteries.

[0004] It considered as the positive active material for lithium secondary batteries which contains iron as a subject, and the lithium content ferric acid ghost of former versatility has been proposed. For example, the lithium iron multiple oxide (143 for example, J.Electrochem.Soc., 2435 (1996)) (LiFeO2) which has tunnel structure or a stratified zigzag structure, the olivine mold LiFePO4 (J. 144 Electrochem.Soc., 1609 (1997)), LiFeO2 (for example, publication number 10–67519) which has hexagonal stratified rock salt type structure are mentioned further. [0005]

[Problem(s) to be Solved by the Invention] Although LiFeO2 which has the above-mentioned tunnel structure or a stratified zigzag structure has a high charge-and-discharge capacity (150 or more mAh/g) which endures LiCoO2 in early stages, the discharge capacity of . olivine mold LiFePO4 which discharge capacity falls to 80% or less of initial capacity by life test of 10 cycle, and has the trouble that a charge-and-discharge cycle property is low is 140 or less mAh/g, and is inadequate as a cell active material. On the other hand, the technical problem that a charge-and-discharge cycle property is very low (10 or less mAh/g) still lower has charge-and-discharge capacity as LiFeO2 which has hexagonal stratified rock salt type structure is shown by J.Electrochem.Soc., 144,

and L177 (1997).

[0006] therefore, a high discharge capacity of 150 or more mAh/g of the former — having — in addition — and the lithium iron multiple oxide excellent in the charge—and–discharge cycle property is not obtained.

[0007] On the other hand, it is known that it is effective in the improvement in a utilization factor of MnO2 which is alkaline cell positive active material to mix metallic oxides, such as Bi2O3 and PbO, (H. 238 S.Wroblowa and N.Gupta, J.Electroanal.Chem., 93 (1987)). Catalyst nature is pointed out as an operation of a metallic oxide in recent years (DeYanG Qu, J.Appl.Electrochem., 29,511 (1999)). It is not clear to mixing with Bi2O3 and PbO being effective in manganese what kind of metallic oxide brings a good operation to other transition metals, for example, iron.

[0008] The place which makes this invention the purpose paying attention to the addition effectiveness of this metallic oxide is to offer the iron content positive active material for nonaqueous electrolyte secondary which has a high charge—and—discharge capacity and has a still better charge—and—discharge cycle property.

[0009]

[Means for Solving the Problem] The alkali-metal content ferric acid ghost and banazin san ghost which have hexagonal stratified rock salt type structure are mixed, and positive active material for nonaqueous electrolyte rechargeable batteries of this invention is characterized by said banazin san ghost being supported by a part of front face [at least] of said alkali-metal content ferric acid ghost.

[0010] Furthermore, in the positive active material for nonaqueous electrolyte rechargeable batteries of this invention, it is characterized by said alkali-metal content ferric acid ghost being alpha-NaFeO 2.

[0011]

5

[Embodiment of the Invention] By making a part of front face [at least] of the alkali-metal content ferric acid ghost which has hexagonal stratified rock salt type structure support a banazin san ghost with this invention, the utilization factor as positive active material for nonaqueous electrolyte rechargeable batteries of an alkali-metal content ferric acid ghost is made to increase sharply, and a cycle property can be made good.

[0012] LiFeO2 and a-NaFeO2 can be used for the alkali-metal content ferric acid ghost of this invention, and Li1-xNaxFeO2 (0 < x < 1) which permuted a part of Na by Li, and LiM1-yFeyO2 (M=Co, Mn, nickel) (0 < y < 1) which permuted iron [some] by other transition-metals elements can also be used for it. However, it is characterized by all having hexagonal stratified rock salt type structure. [0013] It is possible to use V2O3, V2O4, and V2O5 for the banazin san ghost used by this invention. [0014]

[Example] the following — this invention — the example of the positive active material for nonaqueous electrolyte rechargeable batteries is explained. However, this invention is not limited to the following examples.

[0015] 0.03-mol weighing capacity of [example 1] ferric oxide (a-Fe 2O3) and the sodium peroxide (Na 2O2) was carried out, respectively, and after mixing and pelletizing with a mortar, it calcinated at 550 degrees C under the oxygen ambient atmosphere for 20 hours. a-NaFeO2 which has hexagonal stratified rock salt type structure was obtained by continuing, grinding a sample and calcinating at 550 degrees C again for 20 hours. All of the weighing capacity of a sample and mixing with a mortar were performed in the glove compartment under an argon ambient atmosphere.

[0016] The positive active material with which the banazin san ghost was supported by the front face of a-NaFeO2 was produced by carrying out wet blending of a-NaFeO 2 andV2O5 obtained above in a methanol next, so that a mole ratio may be set to 40:1, and drying at 80 degrees C. The acetylene black 20 weight section as an electric conduction agent and the poly hooker vinylidene

(PVDF) 5 weight section as a binder were added to the mixture 75 weight section of a-NaFeO 2 andV2O5 as positive active material, wet blending was carried out to N-MECHIRU 2 pyrrolidone which is a solvent, and it was made the slurry. After making this slurry into both sides of the aluminum mesh which is a charge collector with **, pressing was carried out by 1 t/cm2, it dried at 230 degrees C under the vacuum, and the magnitude 15mmx15mmx0.5mm this invention positive-electrode plate (A1) was produced.

[0017] a-NaFeO 2 and V2O5 which may be based on the [example 2] example 1 was mixed so that a mole ratio might be set to 40:2, and this invention positive-electrode plate (A2) was produced like the example 1 except having considered as positive active material.

[0018] a-NaFeO 2 and V2O5 which may be based on the [example 3] example 1 was mixed so that a mole ratio might be set to 40:4, and this invention positive-electrode plate (A3) was produced like the example 1 except having considered as positive active material.

[0019] a-NaFeO 2 and V2O5 which may be based on the [example 4] example 1 was mixed so that a mole ratio might be set to 40:10, and this invention positive-electrode plate (A4) was produced like the example 1 except having considered as positive active material.

[0020] a-NaFeO 2 and V2O5 which may be based on the [example 5] example 1 was mixed so that a mole ratio might be set to 40:20, and this invention positive-electrode plate (A5) was produced like the example 1 except having considered as positive active material.

[0021] a-NaFeO 2 and V2O5 which may be based on the [example 6] example 1 was mixed so that a mole ratio might be set to 40:40, and this invention positive-electrode plate (A6) was produced like the example 1 except having considered as positive active material.

[0022] a-NaFeO 2 and V2O5 which may be based on the [example 7] example 1 was mixed so that a mole ratio might be set to 40:60, and this invention positive-electrode plate (A7) was produced like the example 1 except having considered as positive active material.

[0023] a-NaFeO 2 and V2O5 which may be based on the [example 8] example 1 was mixed so that a mole ratio might be set to 40:80, and this invention positive-electrode plate (A8) was produced like the example 1 except having considered as positive active material.

[0024] a-NaFeO 2 and V2O5 which may be based on the [example 9] example 1 was mixed so that a mole ratio might be set to 40:100, and this invention positive-electrode plate (A9) was produced like the example 1 except having considered as positive active material.

[0025] a-NaFeO 2 andV2O5 which may be based on the [example 10] example 1 was mixed so that a mole ratio might be set to 40:120, and this invention positive-electrode plate (A10) was produced like the example 1 except having considered as positive active material.

[0026] The comparison positive-electrode plate (B1) was produced like the example 1 except having used a-NaFeO2 independently as [example 1 of comparison] positive active material.

[0027] [Charge-and-discharge property] this invention positive-electrode plate (A1) – (A10) and a comparison positive-electrode plate (B1) were used as the trial pole base material, respectively, and the cel for an experiment was constituted. It is 1 mol/l to a lithium metal and nonaqueous electrolyte in a counter electrode and a reference pole. The mixed solution of the volume ratio 1:1 of ethylene carbonate and diethyl carbonate in which lithium perchlorate was dissolved was used.

[0028] The positive-electrode charge-and-discharge property was investigated using the above-mentioned cel for an experiment. About this invention positive-electrode plate (A1) – (A10) and a comparison positive-electrode plate (B1), after discharging to 1.5V by current density 2 mA/g, it charged to 3.5V by clinch 2 mA/g. When setting discharge capacity with C1 and setting charge capacity with C2, charge-and-discharge cycle efficiency was computed from the following formulas. Charge-and-discharge cycle efficiency (%) =(C2/C1) x100, in addition a charge and discharge test were started from discharge (lithium insertion).

[0029] When setting the mole ratio of a-NaFeO 2 and V2O5 to x and y, the relation between the

value of (y/x) of this invention positive-electrode plate (A1) – (A10) and a comparison positive-electrode plate (B1) and the charge capacity in each cycle is shown in Table 1. Moreover, the cycle efficiency in each cycle of this invention positive-electrode plate (A1) – (A10) and a comparison positive-electrode plate (B1) is shown in Table 2. Furthermore, the charge-and-discharge curve in 3 cycle eye of this invention positive-electrode plate (A6) and a comparison positive-electrode plate (B1) is shown in <u>drawing 1</u> and <u>drawing 2</u>, respectively. [0030]

[Table 1]

正極板		充電容量(mAh/g)		
	(y/x)	1サイクル	2サイクル	5サイクル
A 1	0.025	7	7	7
A 2	0.05	15	16	16
A 3	0. 1	2 7	2 7	2 7
A4 .	0.25	5 2	6 5	7 1
A 5	0.5	9 2	9 6	98
A 6	1. 0	134	134	135
A 7	1. 5	175	170	168
A 8	2. 0	203	191	188
A 9	2. 5	225	208	204
A 1 0	3. 0	239	217	213
B 1	_	3	4	4

[0031] [Table 2]

正極板		サイクル効率 (%)		
	(y/x)	1サイクル	2サイクル	5サイクル
A 1	0.025	8 7	8 9	9 5
A 2	0.05	8 6	94	9 7
A 3_	0.1	8 5	9 4	98
A 4	0.25	86	98	99
A 5	0.5	9 2	96	98
A 6	1. 0	8.8	96	97
A 7	1. 5	8 6	9 7	98
A 8	2. 0	80	99	9 9
A 9	2. 5	79	98	98
A 1 0	3. 0	7 7	98	9 9
B 1		5 6	78	9 2

[0032] Drawing 1 and drawing 2 show that the charge-and-discharge curves differ greatly as compared with a-NaFeO2 (B1) with the mixture (A6) of a-NaFeO 2 andV2O5. namely, in a-NaFeO2 (B1), potential changed gently-sloping — receiving — the mixture (A6) of a-NaFeO 2 andV2O5 — about 2 — the surface smoothness of potential was checked by V. Although the surface smoothness of the potential of about 2 V appears similarly in the initial cycle at the time of using V2O5 as an active material, in future cycles, structure makes it amorphous and it is reported that potential changes gently-sloping (Koshiba et al., DENKI KAGAKU, 332 (1994)). Moreover, if it assumes that only V2O 5 have contributed to the reaction in this invention positive-electrode plate and the capacity (2.1V-1.5V) in an initial cycle discharge process is measured, by V2O5, it will become about 200 mAh/g to becoming 240 mAh/g with this invention positive-electrode plate (A4) (Koshiba et al., DENKI KAGAKU, and 332 (1994)).

[0033] The above result, with this invention positive-electrode plate, the property independent [20V5] has not appeared and it is thought that a-NaFeO2 has contributed to the charge-and-discharge reaction greatly. It is thought that that the capacity of this invention

positive-electrode plate increased compared with the comparison positive-electrode plate (B1) originates in improvement in the utilization factor of a-NaFeO2 by mixing with V2O5. [0034] Starting divalent [of Fe] / trivalent oxidation reduction reaction by about 2 V is reported (KAmine et.al., J.Power.Sources, 81-82,221 (1999)). Therefore, the surface smoothness (<u>drawing 1</u>) of the potential of about 2 V which appeared with this invention positive-electrode plate originates in divalent [of Fe accompanying insertion and desorption of the lithium to a-NaFeO2] / trivalent oxidation reduction reaction, and is considered to have been brought by improvement in the utilization factor of a-NaFeO2.

[0035] Table 1 shows that this invention positive active material whose values of (y/x) are 1-3.0 has a big discharge capacity (130 - 200 mAh/g). Moreover, Table 2 shows that this invention positive active material whose values of (y/x) are 0-2.5 is excellent in a cycle property compared with a comparison positive-electrode plate. Therefore, it turns out that it is desirable to use the mixture of a-NaFeO 2 andV2O5 whose values of (y/x) are 1-2.5 when obtaining the nonaqueous electrolyte positive active material which has a good cycle property by high capacity.

[0036] Although a-NaFeO2 was chosen as the alkali-metal content ferric acid ghost, V2O5 were chosen as the banazin san ghost and this example explained, in a-NaFeOV2O3, the active material which mixed [2] V2O4, and the active material which mixed V2O3, V2O4, and V2O5 to LiFeO2 and FeOOH, the good cycle property was similarly acquired by high capacity.

[0037] By this invention, by high capacity, it succeeds in obtaining the alkali-metal content ferric acid ghost excellent in the charge-and-discharge property for the first time, and it is cheap and contributes to development of the small positive active material for nonaqueous electrolyte rechargeable batteries of an environmental load greatly.

[0038]

[Effect of the Invention] The positive active material for nonaqueous electrolyte rechargeable batteries of this invention It is characterized by supporting a banazin san ghost on a part of front face [at least] of an alkali-metal content ferric acid ghost. If the mole ratio of a-NaFeO 2 andV2O5 is set to x and y when a-NaFeO2 is used for an alkali-metal content ferric acid ghost and V2O5 are used for a banazin san ghost 0 < (y/x) < 3 and when it is referred to as 1 < (y/x) < 2.5 still more preferably, it is high capacity and a cycle property becomes good especially.

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DESCRIPTION OF DRAWINGS

[Brief Description of the Drawings]

[Drawing 1] It is drawing showing the charge-and-discharge property in 3 cycle eye of this invention positive-electrode plate (A6).

[Drawing 2] It is drawing showing the charge-and-discharge property in 3 cycle eye of a comparison positive-electrode plate (B1).

[Translation done.]